Theoretical Study of Solid Materials for CO₂ Capture Technology

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ABSTRACT

Since the current technologies for capturing CO₂ are still too energy intensive, to develop new materials that can capture CO₂ reversibly with acceptable energy costs are needed. By combining thermodynamic database mining with first principles density functional theory and phonon lattice dynamics calculations, a theoretical screening methodology to identify the most promising CO₂ sorbent candidates from the vast array of possible solid materials have been proposed and validated at the National Energy Technology Laboratory (NETL). The advantage of this method is that it identifies the thermodynamic properties of the CO₂ capture reaction as a function of temperature and pressure without any experimental input beyond crystallographic structural information of the solid phases involved.

Keywords: CO₂ capture, solid sorbents, *ab initio* thermodynamics

1 INTRODUCTION

Carbon dioxide is one of the major combustion products which once released into the air can contribute to global climate change.[1-6] It is generally accepted that current technologies for capturing CO₂ are still too energy intensive and thus cost prohibitive for implementation on coal based power plants. Hence, there is a critical need for development of new materials that can capture CO₂ reversibly with acceptable energy costs. Accordingly, solid sorbents have been reported in several previous studies to be promising candidates for CO₂ sorbent applications through a reversible chemical transformation due to their high CO₂ absorption capacities at moderate working temperatures.[3, 7, 8] Solid sorbents containing alkali and alkaline earth metals have been reported in several previous studies to be promising candidates for CO2 sorbent applications due to their high absorption capacity at moderate temperatures.[9-11]

During past few years we developed a theoretical methodology to identify promising solid sorbent candidates for CO₂ capture by combining thermodynamic database searching with *ab initio* thermodynamics obtained based on first-principles density functional theory (DFT) and lattice phonon dynamics.[9-18] As shown in Fig.1, the primary outcome of our screening scheme is a list of promising CO₂ sorbents with optimal energy usage.

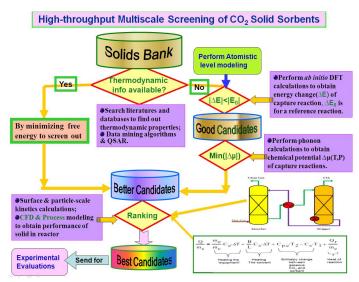


Fig. 1. Schematic of our screening methodology.

2 CALCULATION METHODS

The complete description of the computational methodology together with relevant applications can be found in our previous publications.[9-16, 19] The CO₂ capture reactions of solids can be expressed generically in the form (for convenient description, we normalized the reaction to 1 mole of CO₂)

$$\sum_{Ri} n_{Ri} Solid _R_i + CO_2 \leftrightarrow \sum_{Pi} n_{Pi} Solid _P_j$$
 (a)

where n_{Ri} , n_{Pj} are the numbers of moles of reactants (R_i) and products (P_j) involved in the capture reactions. We treat the gas phase CO_2 as an ideal gas. By assuming that the difference between the chemical potentials $(\Delta \mu^0)$ of the solid phases of reactants (R_i) and products (P_j) can be approximated by the difference in their total energies (ΔE^{DFT}) , obtained directly from DFT calculations, and their vibrational free energies of phonons dynamics and by ignoring the PV contribution terms for solids, the variation of the Gibbs free energy (ΔG) for reaction (a) with temperature and CO_2 pressure can be written as [6, 9-16, 19]

$$\Delta G (T, P) = \Delta \mu^{0} (T) - RT \ln \frac{P_{CO_{2}}}{P_{0}}$$
 (1)

where

$$\Delta \mu^{0}(T) \approx \Delta E^{DFT} + \Delta E_{ZP} + \Delta F^{PH}(T) - G_{CO_{3}}^{0}(T)$$
 (2)

Here, ΔE^{DFT} is the DFT energy difference between the reactants and products of the reaction (a), ΔE_{ZP} is the zero point energy difference between the reactants and products and can be obtained directly from phonon calculations. ΔF^{PH}

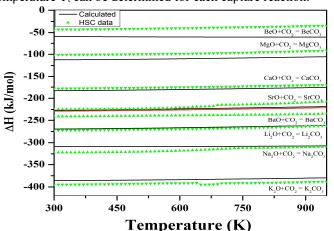
is the phonon free energy change excluding zero-point energy (which is already counted into the ΔE_{ZP} term) between the solids of products and reactants. $P_{\rm CO2}$ is the partial pressure of CO_2 in the gas phase and P_0 is the standard state reference pressure taken to be 1 bar. The heat of reaction $(\Delta H^{\rm cal}(T))$ can be evaluated through the following equation

$$\Delta H^{cal}(T) = \Delta \mu^{0}(T) + T[\Delta S_{PH}(T) - S_{CO_{T}}(T)]$$
(3)

where $\Delta S_{PH}(T)$ is the difference of entropies between product solids and reactant solids. The free energy of CO_2 (G^0_{CO2}) can be obtained from standard statistical mechanics,[10, 11, 16] and its entropy (S_{CO2}) can be found in the empirical thermodynamic databases.[20] Equation (1) provides the relationships of Gibbs free energy change of reaction (a) versus temperature and CO_2 pressure. Obviously, when set ΔG =0, the P-T relationship (van't Hoff plot) is obtained to determine the turnover temperature T_t :

$$T_{t} = \frac{\Delta \mu^{0}(T)}{R \ln \frac{P_{CO_{2}}}{P_{0}}}$$
 (4)

Based on it, at giving CO_2 pressure P_{CO_2} , the turnover temperature T_t can be determined for each capture reaction.



3 RESULTS AND DISCUSSION

3.1 Method Validation: Applications to alkali and alkaline earth metal oxides and hydroxides

As an example, Figure 2 shows the calculated and experimental measured thermodynamic properties of the reactions for alkali and alkaline earth metal oxides capture of CO₂. From it, one can see that, except for BeO+CO₂→BeCO₃ reaction, overall, the calculated results are in good agreement with HSC experimental data. These findings indicate that our theoretical approach can predict the right thermodynamic properties of various solid reacting with CO₂ if the right crystal structure is known or is easy to be determined. The larger discrepancy observed for BeO/BeCO₃ system is due to lack of the crystal structure information of BeCO₃. As the only one input property of the solid in the *ab initio* thermodynamics calculations, this indicates that in order to obtain reliable results, the crystal structure must be known or can be easily predicted correctly.

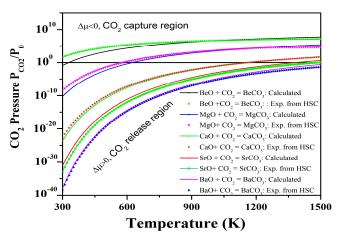


Fig. 2. (left)The calculated heats of reaction as a function of temperature for eight metal oxides. The solid lines were computed from DFT including phonon contributions. The dashed lines were computed from the HSC package. The discontinuities in the HSC curves indicate phase transitions; (right) The calculated chemical potentials (Gibbs free energy) versus CO₂ pressures (in logarithm scale) and temperatures for the reactions of oxides capturing CO₂ without water involved.

3.2 Mixing Solid Sorbents to Adjust Operating Temperature

For a given CO_2 capture process, its optimal working conditions (CO_2 pressures of pre- and after-capture, absorption/desorption temperature range (ΔT_o), *etc.*) were fixed. However, at a given CO_2 pressure, the turnover temperature (T_t) of an individual solid capture CO_2 reaction is fixed. Such T_t may be outside the operating temperature range ΔT_o for a particularly capture technology. In order to adjust T_t to fit the practical working through reversible chemical transformations, the chemical properties of solids must be modified to change the $\Delta G(T, P)$ in Eq. (1). If we want to increase the T_t to high temperature range, the $\Delta G(T, P)$

P) should be more negative. To achieve it, we can either destabilize the reactants (sorbents) or stabilize the products or do both. On contrary, if we want to decrease T_t to low temperature range, the $\Delta G(T, P)$ should be less negative, which can either stabilize the reactant (sorbents) or destabilize the products or do both. In other words, mixing stabilizer/destabilizer in solid could change thermodynamic properties of their CO₂ capture reactions to shift T_t. Some mixing examples are given in Table 1. As one can see that the mixed sorbent could be a new formed solid (e.g. lithium silicates) or just simple mixture (e.g. MgO+Na₂CO₃) to change the chemical properties of reactants and products. Depending on the main captor A and the direction of T_t shifts, different mixed solid B and mixing

ratio could be determined. Although one can mix any number of solids to form a new CO₂ sorbent, in order to focus

exploring the nature of mixtures; here we restrict ourselves with cases of two and three mixed solids.

Table 1. Mixing schemes and examples of mixed solids and their effects on CO₂ capture

Main captor		New formed sorbent	Examples of CO ₂ capture	Effect on CO ₂ capture capacity & operating	Refs.
A	(one or more)	C	reaction	T	
N ₂ O, N=Li,	SiO2, TiO2, ZrO2,	Lithium silicates,	$Li_4SiO_4 + CO_2 = Li_2CO_3$	(1) B stabilizes A, but does not capture CO ₂ ;	[[10,
Na, K	Al ₂ O ₃ , MeO (Me:	lithium zirconates,	$+ Li_2SiO_3$	(2) Compared to pure N2O, the T _t shifts to	12, 19,
MO, M=Mg,	transition metal)	calcium aluminum	$Li_2ZrO_3 + CO_2 = Li_2CO_3$	low-T range;	21-23]]
Ca	Two solids	oxides	$+ ZrO_2$	(3) The maximum CO ₂ capture capacity	
	mixing with		$^{1}/_{12}Ca_{12}Al_{14}O_{33}+CO_{2}=$	decreases with increasing solid B.	
	different ratios		$CaCO_3 + \frac{7}{12}Al_2O_3$		
MO, M=	N ₂ O or N ₂ CO ₃ ,	MO+N ₂ O	$MgO+Na_2CO_3+CO_2 =$	(1) N ₂ O (or N ₂ CO ₃) stabilizes the product to	[[24-26]]
Mg, Ca	N= Li, Na, K	MO+N ₂ CO ₃	$Na_2Mg(CO_3)_2$	form double salt and does not capture CO ₂	
		MgO+CaCO ₃	$MgO+CaCO_3+CO_2 =$	in the ΔT_0 range;	
	Two solids		$MgCa(CO_3)_2$	(2) Compared to pure MO, the T _t shifts to	
	mixing with			high-T range;	
	different ratios			(3) The maximum CO ₂ capture capacity	
				decreases with B.	
N_2O , $N=Li$,	SiO ₂ +MeO	Li ₂ MeSiO ₄ (Me= Fe,	$Li_2MeSiO_4 + CO_2 =$	(1) B stabilizes both A and product. The final	
Na, K		Co, Ni, Mg, Mn, Zn,	$Li_2CO_3 + MeSiO_3$	effect depends on the bonding strength of	
	Three solids	etc.)		MeO with A and SiO ₂ ;	
	mixing with			(2) T _t may shift to low- and high-T range;	
	different ratios			(3) The maximum CO ₂ capture capacity	
				decreases with B;	
N_2O , $N=Li$,	N ₂ O+SiO ₂	Li _{2-x} Na _x ZrO ₃	$Li_{2-x}N_xZrO_3 + CO_2 =$	(1) B stabilizes A and portion of B (N2O) also	[[17]]
Na, K	N_2O+ZrO_2	$Li_{2-x}K_xZrO_3$	$ZrO_2 + {(2-x)/_2}Li_2CO_3 +$	involves in CO ₂ capture;	
	MO+SiO ₂	Li _{4-x} Na _x SiO ₄	$^{x}/_{2}N_{2}CO_{3}$	(2) T _t may shift to low-T range;	
	Three solids	They also can be	$Li_{4-x}N_xSiO_4 + CO_2 = {}^{(2-)}$	(3) The maximum CO ₂ capture capacity	
	mixing with	treated as doped	$x)/_2Li_2CO_3 + x/_2N_2CO_3 +$	decreases with weaker portion of B (SiO ₂ ,	
	different ratios	materials	Li ₂ SiO ₃	ZrO_2 , etc.)	

As shown in Table 1, as effective main CO₂ captors through chemical reactions to form carbonates, alkali and alkaline earth metal oxides are important of interests due to their easy reacting with CO₂ and low costs. The problem is that they can strongly react with CO₂ to form carbonates, but their turnover temperatures T_t are very high and only can be regenerated at very high temperatures which are unsuitable for many CO2 capture technologies. Hence, mixing with other solids to shift their Tt becomes important for their suitability as CO₂ sorbents. Generally, when we mix two solids A and B to form a new sorbent C, the turnover temperature of the newly resulted system (T_C) is located between those of A and B (T_A, T_B) individuals. Here it was assumed that A is a strong CO_2 sorbent while B is a weak CO₂ sorbent and T_A>T_B. Also, we assumed that the desired operating temperature T_O is between T_A and T_B ($T_A > T_O > T_B$). Depending on the properties of A and B, as shown in Table 1 we have typically three scenarios to synthesize the mixing sorbent C, as demonstrated in Fig. 3.

In the case of Li₂O, it is a very strong CO₂ sorbent which forms Li₂CO₃. However, its regeneration from Li₂CO₃ only can occur at very high temperature (T_A). In order to move its T_A to lower temperatures, one can mix some weak CO₂ sorbents (such as SiO₂, ZrO₂). With different mixing ratio of Li₂O/SiO₂(or ZrO₂), different stable lithium silicates (zirconates) can be formed. By performing thermodynamic property calculations of these solids capturing CO₂ reactions,[10, 12, 19, 21-23] the relationship of Gibbs free

energy, P_{CO2} , and T of the CO_2 capture reactions by the mixed Li_2O/SiO_2 and Li_2O/ZrO_2 solids with different mixing ratios can be obtained. Fig. 3(a) shows the turnover temperature and the CO_2 capture capacity of Li_2O/SiO_2 and Li_2O/ZrO_2 mixtures versus the ratio of Li_2O/SiO_2 or Li_2O/ZrO_2 .[9-12, 19, 21, 22] When the SiO_2/Li_2O or ZrO_2/Li_2O ratios are increased by adding more SiO_2 or ZrO_2 into Li_2O , the turnover temperatures of mixed sorbents (T_t) are shifted more to lower temperature range.

Opposite to previous case, in this case, we want to increase turnover temperature (T_t). Since T_B is lower than T_O , by mixing A into B will increase the turnover temperature T_C of the C solid to values closer to T_O . In this way, the function of A is either to destabilize solid A or to stabilize the captured products. For example, pure MgO has a very high theoretical CO_2 capture capacity. However, its turnover temperature (250 °C) is lower than the required temperature range of 300-470 °C used in warm gas clean up technology and its practical CO_2 capacity is very low, and therefore, it cannot be used directly as a CO_2 sorbent in this technology.[25-27] Fig. 3(b) shows an example of mixing Na_2O (or Na_2CO_3) with MgO to shift the T_t of MgO to higher temperature range.

Another case is that we want both A and B components are active to capture CO_2 , and the CO_2 capacity of the mixture is the summation of those of A and B. As we know another potential advantage of mixing solids is to increase the surface area of the solids in order to have faster reaction rates. Such a mixing scenario doesn't show too much

advantage in shifting the capture temperature, but may enhance the kinetics of the capture process and eventually make the mixtures more efficient. Fig. 3(c) shows a case of Li-/K- doped Na₂ZrO₃ capture CO₂ through the following reaction (b):[17, 28]

$$M_x N_{2-x} Zr Q_3 + C Q_2 \leftrightarrow \frac{x}{2} M_2 C Q_3 + \frac{2-x}{2} N_2 C Q_3 + Zr Q_2$$
 (M,N=Li,Na,K) (b)

Compared to pure Na₂ZrO₃, overall, the Li- and Kdoped mixtures Na_{2-x}M_xZrO₃ have lower turnover temperatures (T_t). The calculated results show that the shift in T_t depends not only on the doping element, but also depends on the doping level.

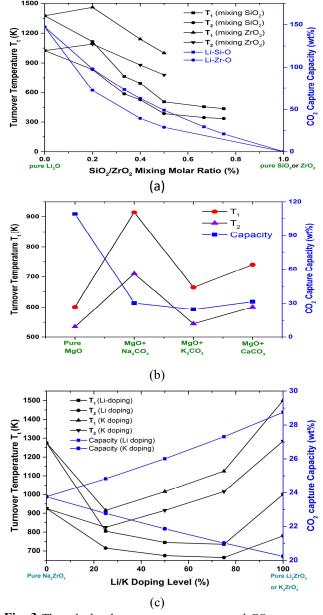


Fig. 3 The calculated turnover temperatures and CO₂ capture capacity versus mixed/doped solids. T₁ are the turnover temperatures under pre-combustion conditions with CO₂ partial pressure at 20 bars, while T₂ are the turnover temperatures under post-combustion conditions with CO₂ partial pressure at 0.1 bar. (a) Li₂O/SiO₂(ZrO₂) mixing systems, (b) MgO/XCO₃(X=Na, K,

Ca) mixing systems, (c) Na_xM_{2-x}ZrO₃ and M₂ZrO₃(M=Li, Na, K, x=0, 0.5, 1.0, 1.5, 2.0) doping systems

4 CONCLUSIONS

The obtained results can be used to provide insights for designing new CO₂ sorbents. Therefore, although one single material taken in isolation might not be an optimal CO2 sorbent to fit the particular needs to operate at specific temperature and pressure conditions, by mixing or doping two or more materials to form a new solid, the calculated results showed that it is possible to synthesize new CO₂ sorbent formulations which can fit the industrial needs. Our results also show that computational modeling can play a decisive role for identifying materials with optimal performance.[17, 18, 29-37]

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